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## Absolute Rate Constants in Free-Radical Polymerization. III.\* Determination of Propagation and Termination Rate Constants for Styrene and Methyl Methacrylate

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#### ABSTRACT

A spatially intermittent polymerization (SIP) reactor has been used for determination of absolute rate constants in photoinitiated, free-radical polymerization of styrene (STY) and methyl methacrylate (MMA). Experimental data are reported in the temperature range  $15-30^{\circ}$ C and in the high molecular weight region for MMA and STY. Additional experimental data are reported at  $30^{\circ}$ C and various lower molecular weights for STY which indicate that the propagation rate constant K<sub>n</sub> is independent of polymer molecular weight, and K<sub>t</sub>

is dependent on molecular weight, especially at low molecular weight, approaching an approximately constant value at high molecular weight.

<sup>\*</sup>For Part II see Mahabadi and O'Driscoll [1].

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#### INTRODUCTION

In a recent paper [1] we derived an expression for the rate constant of the termination reaction between two radicals,  $K_{+}$ , which

permits an a priori calculation of it based on experimentally accessible physical parameters. We have shown that  $K_t$  may be represented by

$$K_{t} = F_{1}(\zeta, T) F_{2}(\alpha, N)$$
(1)

where  $F_1$  is a function only of temperature and the segmental frictional coefficient  $\zeta$ , and  $F_2$  is a function of the chain expansion coefficient  $\alpha$  of the reacting polymer radicals and of their chain length N. The value of  $F_2$  for reacting polymer radicals A and B is given by Eq. (2):

$$F_{2} = (\alpha_{A}\alpha_{B})^{-1} \cdot {}^{3} \{ 1 - 0.37 (\alpha_{A}\alpha_{B})^{-0.37}$$

$$[ 1 - (32N_{0}/3\pi)^{1/2} (N_{A}N_{B})^{-1/4}/(\alpha_{A}\alpha_{B})^{1/2} ] \}$$
(2)

where  $N_0$  is the number of monomer units in a segment.

The experimentally determined values of the termination rate constant  $\overline{K}_t$  in free-radical polymerization is the average rate constant for reaction between all possible pairs of radicals existing in the radical population and can be expressed [2] by Eq. (3).

$$\overline{\mathbf{K}}_{t} = \frac{\sum_{\mathbf{N}_{A}} \sum_{\mathbf{N}_{B}} \mathbf{K}_{t} (\mathbf{N}_{A}, \mathbf{N}_{B}) \mathbf{P}_{\mathbf{N}_{A}} \mathbf{P}_{\mathbf{N}_{B}}}{\sum_{\mathbf{N}_{A}} \sum_{\mathbf{N}_{B}} \mathbf{P}_{\mathbf{N}_{A}} \mathbf{P}_{\mathbf{N}_{B}}}$$
(3)

where  $P_{N_A}$  is the concentration of radicals of size  $N_A$  and  $K_t(N_A, N_B)$  is the actual value of  $K_t$  for radicals of length  $N_A$  and  $N_B$ .

In free-radical polymerization the deviation between the measured value of  $\overline{K}_{t}$  (at a given average degree of polymerization,  $\overline{X}_{n}$ ) and

 $K_t(N_A, N_B)$  when  $N_A = N_B = \overline{X}_n$  can be significant when  $K_t$  is a strong function of chain length and/or when the polydispersity ratio  $\overline{X}_w/\overline{X}_n$ is high. We have found [1] that letting  $N_A = \overline{X}_n = N_B/2$  serves as a good approximation in Eq. (2) for calculation of  $\overline{K}_t$ . Equation (2) and existing experimental data [3, 4] indicate that  $K_t$  for a given polymer-solvent system decreases as the chain length increases and approaches a constant value at high chain length. Therefore, for a polymerization which is carried out at high  $\overline{X}_n$  ( $\overline{X}_n > 1000$ ),  $K_t(N_A, N_B)$ can be equated to the experimentally measurable  $\overline{K}_t$ . Therefore Eq. (1) for the high molecular weight region can be written as

$$\widetilde{\mathbf{K}}_{t} = \mathbf{F}_{1}\mathbf{F}_{2} \qquad \qquad \text{For } \mathbf{N} > 1000 \qquad (4)$$

In Eq. (4),  $F_2$  is nearly constant, being only a weak function of molecular weight for N > 1000 and can be calculated from Eq. (2).

Equation (4) permits the estimation of  $F_1$  by using the calculated value of  $F_2$  and the experimentally measured value of  $\overline{K}_t$  at a high, measured molecular weight.

The values of  $F_1$  can be obtained at various temperatures and then can be used along with Eqs. (1) and (2) to predict  $K_t$  for a given polymer-solvent system at any temperature and for any chain length

of reacting polymer radicals. In a previous paper [3] we used a spatially intermittent polymerization (SIP) reactor to determine the propagation rate constant  $K_p$  and average termination rate constant  $\overline{K}_t$  in the photoinitiated polymerization of methyl methacrylate (MMA) at 25°C. We found that  $\overline{K}_t$  in the bulk polymerization of MMA at 25°C is a strong function of chain length at low molecular weight and a much weaker function at  $\overline{X}_p > 1000$ .

The  $K_p$  and  $\overline{K}_t$  values reported below have been obtained in the nonstationary state, for low conversion polymerization, utilizing [3] Eqs. (5) and (6)

$$K_{p} = \frac{\overline{R}_{p}/R_{ps}}{2\tau_{s}M[(\overline{X}_{n})^{-1} - C_{try}(y/M)]}$$
(5)

$$\overline{K}_{t} = \frac{\overline{R}_{p}/R_{ps}^{2}}{2\tau_{s}^{2} [(\overline{X}_{n})^{-1} - C_{try}(y/M)]}$$
(6)

where  $\overline{R}_{p}$  and  $\overline{X}_{n}$  are the average rate and the average degree of polymerization for the nonstationary state condition,  $R_{ps}$  is the rate for stationary-state polymerization, M is monomer concentration, and  $C_{try}$  is the constant for chain transfer to species Y.  $\tau_{s}$  is the mean radical lifetime under steady illumination. This method for determination of  $\tau_{s}$ , by analyzing the nonstationary-state data, has been described in detail [3]. Equations (5) and (6) have been derived based on the assumption that the termination is by combination and the average square of the radical concentration is equal to the square of the average radical concentration:

$$(\mathbf{P}\cdot)^2 = (\overline{\mathbf{P}}\cdot)^2 \tag{7}$$

If a fraction X of the termination occurs by disproportionation, then the rate constants in Eqs. (5) and (6) must be multiplied by the quantity (1 + X).

In this paper,  $K_p$  and  $\overline{K}_t$  for bulk polymerization of STY and MMA at high molecular weight and various temperature are reported. The

temperature dependence of  $K_p$  and  $\overline{K}_t$  for STY and MMA are presented.

#### EXPERIMENTAL

#### Materials

Monomer, purified by distillation is mixed with initiator after deaeration by several freeze-thaw cycles. The monomer-initiator solution is stored in the dark under nitrogen at 0°C in a vessel which serves as a reservoir for a syringe sump (SIP reactor feed pump, sage model 335). Initiator, azobisisobutyronitrile (AIBN), was recrystallized from acetone, dried under vacuum at room temperature, and stored in the dark in a freezer.

#### Apparatus and Procedure

Apparatus (SIP reactor) and procedure have been previously described [3]. The rate of polymerization was obtained by measuring the concentration of polymer in the effluent from the SIP reactor by using a chromatographic column packed with deactivated 75-125  $\mu$ m Porasil particles. The molecular weight of the polymer in the SIP effluent was determined by using a Waters Associates, Model ALC-301 gel-permeation chromatograph. Necessary values were taken from the literature for monomer densities [5, 6] and C<sub>t,m</sub> [7, 8] as a function of temperature.

#### RESULTS

The kinetic studies were carried out for initiator concentrations between 0.015 and 0.075 mole/liter and at temperatures of  $15-30^{\circ}$ C. All of the data were obtained for conversions less than 1%.

The values of  $K_p$  and  $\overline{K}_t$  obtained by the procedure based on Eqs. (5) and (6) for STY and MMA are given in Tables 1 and 2, respectively. It is clear that, for styrene at 30°C,  $K_p$  is independent of  $\overline{X}_n$  over a wide range, while  $\overline{K}_t$  is a strong function of  $\overline{X}_n$  at low molecular weight and a much weaker function at  $\overline{X}_n > 2000$ . An identical situation was previously established for MMA [3]. Values of  $K_p$  and  $\overline{K}_t$  from Tables 1 and 2 are plotted against 1/T

in Fig. 1; for STY, only the data for  $\overline{X}_n = 2400$  were used. The Arrhenius relationships calculated by least-square analysis are as follows (RT = cal/mole):

MMA 
$$(\overline{X}_{n} = 10^{4})$$
:  
 $K_{p} = 4.92 \times 10^{5} \exp \{-4353/RT\}$ 
(8)  
 $\overline{K}_{t} = 9.80 \times 10^{7} \exp \{-701/RT\}$ 
(9)

STY 
$$(\overline{X}_n = 2.5 \times 10^3)$$
:

$$K_{\rm p} = 1.09 \times 10^7 \exp\{-7051/RT\}$$
 (10)

$$\overline{K}_{+} = 1.703 \times 10^{9} \exp \{-2268/RT\}$$
 (11)

	p L				
Т (°С)	[AIBN] <sub>0</sub> (mole/liter)	x <sub>n</sub>	K <sub>p</sub> (liter/mole-sec)	$\frac{\overline{\mathbf{K}}_{\mathbf{t}} \times 10^{-6}}{(\text{liter/mole-sec})}$	
15	0.075	2705	48.7	32.4	
20	0.045	2686	60.	34.6	
25a	0.03	2600	73.6	37.	
30	0.015	2770	89.4	39.	
30	0.02	2400	89.6	39.4	
30	0.03	1650	89.8	44.2	
30	0.045	1300	89.6	47.	
30	0.075	950	89.7	50.	
30	0.095	780	90.	53.	

TABLE 1. Values of  $K_{n}$  and  $\overline{K}_{n}$  for Styrene

<sup>a</sup>Data of O'Driscoll and Mahabadi [3].

TABLE 2. Values of  $K_p$  and  $\overline{K}_t$  for Methyl Methacrylate

T (°C)	[AIBN] <sub>0</sub> mole/liter	$\overline{\mathbf{X}}_{\mathbf{n}} \times 10^{-3}$	K <sub>p</sub> (liter/mole-sec)	$K_t \times 10^{-6}$ (liter/mole-sec)
15	0.05	14.95	244.1	28.8
<b>2</b> 0	0.04	13.0	278.5	29.4
<b>2</b> 5 <sup>a</sup>	0.03	13.91	315.0	30.
30	0.025	10.62	356.0	30.6

<sup>a</sup>Data of O'Driscoll and Mahabadi [3].

Values of  $\overline{K}_{t}$  for STY at 30°C are plotted as a function of  $\overline{X}_{n}$  in Fig. 2.

#### DISCUSSION

The activation energies in Eqs. (8)-(11) are consistent with those tabulated in the literature [9]. They are expected to be nearly independent of chain length at high  $\overline{X}_n$ . The absolute values of  $K_p$  and  $\overline{K}_t$  are subject



FIG. 1. K<sub>p</sub> for MMA (•) and STY (•) and  $\overline{K}_t$  for MMA ( $\Box$ ) and STY (•) as a function of temperature.

to correction for the occurrence of termination by disproportionation as mentioned above. However, the exact amount of disproportionation, if any, must be regarded as unknown at present since previous workers, who have dealt with this, have not acknowledged the existence of a chain-length dependence of  $K_r$ . It is probable that styrene terminates

exclusively by combination while MMA has significant combination [10]. Because the  $\overline{K}_{t}$  values in Eqs. (9) and (11) are for moderately high

molecular weight, they may also be used, with minor corrections, for higher molecular weights. Furthermore, from Eq. (2), we can calculate the temperature-independent values of  $F_2$  at the high molecular weights where  $\overline{K}_{t}$  was measured and, using Eq. (4), calculate  $F_1$ ,

at various temperatures.

To utilize Eq. (2) we must have a relation between the chain



FIG. 2.  $\overline{K}_t$  for STY at 30°C as a function of  $\overline{X}_n$ . Solid line calculated from Eqs. (1), (2), and (13) as described in text.

expansion coefficient  $\alpha$  and chain length N. As we have previously shown [1] this is  $\alpha = .857 \text{ N}^{0.0473}$  for poly(MMA) dissolved in its monomer. Using Ito's data [11] we find  $\alpha = 0.831 \text{ N}^{0.067}$  for poly(STY) in its monomer. As before [1], we take N<sub>0</sub> = 4.84 for MMA and 6.0 for STY. Thus we calculate

$$F_1$$
 (MMA) = 3.1 × 10<sup>8</sup> exp {-695/RT} (12)

$$F_1$$
 (STY) = 6.0 × 10<sup>9</sup> exp {-2268/RT} (13)

Equation (13) with Eqs. (1) and (2) now permits us to calculate the solid line shown in Fig. 2 for comparison with experimentally measured  $\overline{K}_t$  values. The agreement is reasonable, but not exact. As noted above, deviation between the experimentally measured  $\overline{K}_t$  and that calculated for  $\overline{X}_n$  is to be expected when  $\overline{K}_t$  is a strong function of shain length

function of chain length.

Some comment should be made concerning the approximation that  $(\overline{\mathbf{P}^{\cdot 2}}) = (\mathbf{P}_{\cdot})^2$  when conversion in SIP reactor is less than 1%; this was used for extimation of  $K_n$  and  $K_t$  in Eq. (5) and (6). The

exact value of the ratio  $(\overline{\mathbf{P}^{\cdot 2}})/(\overline{\mathbf{P}}^{\cdot})^2$  can be calculated by using the average initiation rate in the SIP reactor which can be written as

$$\overline{(\mathbf{P}^{2})}/(\overline{\mathbf{P}})^{2} = [(\mathbf{r}\mathbf{z}^{2} + 1)/(\mathbf{r} + 1)] (\mathbf{R}_{ps}/\overline{\mathbf{R}}_{p})^{2}$$
(7)

where r = 6.98 is the ratio of dark to light length in our SIP reactor, and z is the fraction of thermal polymerization,  $\overline{R}_p/R_{ps}$  is a com-

plex function of r, z, and flow rate [3]. For conversion less than 1% and  $z \le 0.2$  the maximum error of approximating  $(\overline{\mathbf{P} \cdot \mathbf{^2}})$  by  $(\overline{\mathbf{P}} \cdot)^2$  is calculated to be less than 8%.

Since the initiator concentrations were low and monomer concentrations were high (bulk polymerization in the high molecular weight region) the effect of primary radical termination on the estimated values of  $K_p$  and  $\overline{K}_t$  is negligible.

The simultaneous determination of  $\overline{X}_n$  and  $\overline{R}_p$  under intermittent illumination at very low conversion and high molecular weight enables us to see the effect of chain length dependence on  $\overline{K}_t$  and eliminate it

from the derived values of the activation energies of propagation and termination.

The calculation of  $K_{t}$  for MMA or STY by Eq. (1), (2), and (12) or

(13) is possible now for any solvent-polymer system where the solvent viscosity  $\eta_0$  and the dependence of  $\alpha$  on N are known. The latter can be obtained from intrinsic viscosity-molecular weight relations [1]. An a priori estimate of  $F_1$  can be done by using the previously derived [1] relation:

$$\mathbf{F}_{1} = \frac{4}{3} \times 10^{-3} \, \mathrm{N}_{av} \, \mathrm{k}_{B} \, \mathrm{TR/a}_{m} \mathrm{N}_{0} \eta_{0} \, (\zeta_{0}/\zeta_{sp}) \tag{14}$$

Using the data previously given [1] we find at  $25^{\circ}C$ 

 $F_1$  (MMA) = 1.15 × 10<sup>8</sup>

compared to the value of  $0.96 \times 10^8$  from Eq. (12) and

 $F_1$  (STY) = 1.38 × 10<sup>8</sup>

compared to the value of  $1.30 \times 10^8$  from Eq. (13).

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